

How To Do Simple Calculations With Quantum ESPRESSO



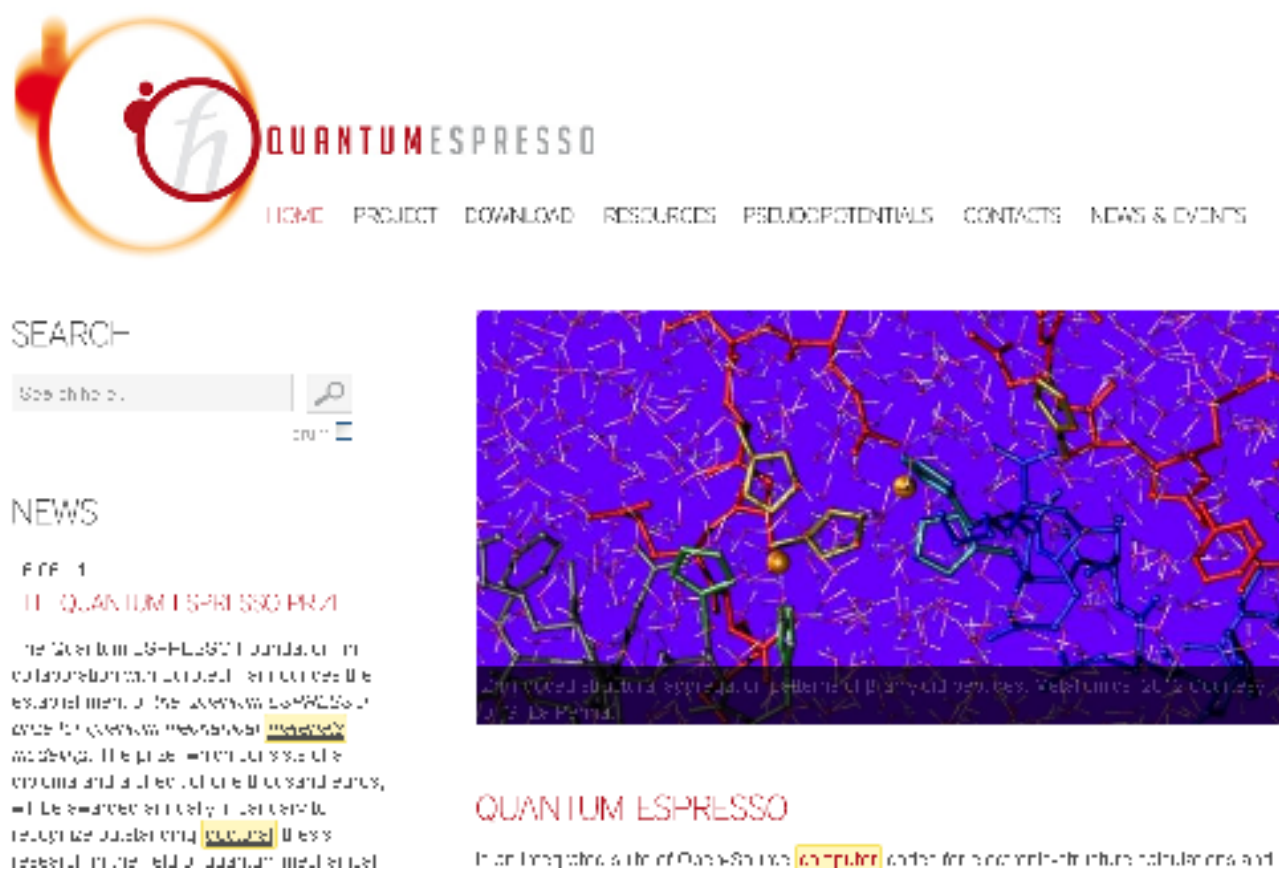
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I. About The Quantum ESPRESSO Distribution

Quantum ESPRESSO

- www.quantum-espresso.org



The screenshot shows the homepage of the Quantum ESPRESSO website. At the top, there is a logo consisting of a stylized 'h' inside a circle, followed by the text 'QUANTUM ESPRESSO'. Below the logo is a navigation bar with links: HOME, PROJECT, DOWNLOAD, RESOURCES, PSEUDOPOTENTIALS, CONTACTS, and NEWS & EVENTS. On the left side, there is a search bar with the placeholder text 'Search here...' and a magnifying glass icon. Below the search bar, there is a 'NEWS' section with the heading 'NEWS' and a sub-heading 'RFP-1'. The text in the RFP-1 section mentions a competition for quantum mechanical modeling. On the right side, there is a large image showing a molecular structure with atoms represented by spheres and bonds by lines. Below the image, there is a caption: 'Chemical structure of a molecule (C10H10N2O2) calculated with Quantum ESPRESSO. (C. J. Pickard)'. At the bottom, there is a section titled 'QUANTUM ESPRESSO' with the text 'For frequent users of Quantum ESPRESSO, contact for help with the software and'.

The Quantum ESPRESSO Software Distribution



The DEMOCRITOS center of Italian INFM is dedicated to atomistic simulations of materials, with a strong emphasis on the development of high-quality scientific software

Quantum ESPRESSO is the result of a DEMOCRITOS initiative, in collaboration with several other institutions (ICTP, CINECA Bologna, EPF Lausanne, Princeton University, MIT, Paris VI, Oxford, IJS Ljubljana,...)

Quantum ESPRESSO is a distribution of software for atomistic simulations based on electronic structure, using density-functional theory (DFT), a plane waves (PW) basis set and pseudopotentials (PP)

Quantum ESPRESSO stands for *Quantum opEn-Source Package for Research in Electronic Structure, Simulation, and Optimization*

P. Giannozzi

Why “Quantum ESPRESSO”?!



Trieste, Tommaseo (ph. G. Crozzoli)



Licence for Quantum ESPRESSO



Quantum ESPRESSO is distributed under the *GNU (Gnu's Not Unix) General Public License (GPL)*, probably the most common free-software license. Basically:

- The source code is available.
- You can do whatever you want with the sources, but if you distribute any derived work, you have to distribute under the GPL the sources of the derived work.

Advantages:

- Everybody – including commercial entities – can contribute.
- Nobody can “steal” the code and give nothing back to the community.

The most successful example is probably the Linux Kernel.

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Quantum ESPRESSO: Organization

The distribution is maintained as a single CVS (Concurrent Version System) tree. Available to everyone anytime via anonymous (read-only) access.

- *Web site:* <http://www.quantum-espresso.org>
- *Wiki:* http://www.quantum-espresso.org/index.php/Main_Page
contains the updated documentation
- *Developers' portal:* <http://www.qe-forge.org>
integrated developer environment, open to external contributions

Mailing lists:

- `pw_users`: used by developers for announcements about Quantum ESPRESSO
- `pw_forum`: for general discussions (all subscribed users can post)

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Quantum ESPRESSO as a distribution

Quantum ESPRESSO aims at becoming a *distribution* of packages, rather than a single, monolithic, tightly integrated package. Main packages:

- PWscf: self-consistent electronic structure, structural relaxation, molecular dynamics
- CP/FPMD: variable-cell Car-Parrinello molecular dynamics

They share a common installation method, input format, PP format, data output format, large parts of the basic code. More packages:

- PHonon: linear-response calculations (phonons, dielectric properties)
- PostProc: graphical and postprocessing utilities (density of states, STM, etc.)
- atomic: pseudopotential generation code
- PWGui: a Graphical User Interface for production of input files

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Quantum ESPRESSO as a distribution



OTHER PACKAGES

WANNIER90: Maximally localized Wannier functions

Pwcond: Ballistic conductance

WanT: Coherent Transport from Maximally Localized Wannier Functions

Xspectra: Calculation of x-ray near edge absorption spectra

QE-GIPAW: EPR and NMR Chemical Shifts


GWL: GW Band Structure

TD-DFT: Time-Dependent Density Functional Pert. Theory

What Can Quantum ESPRESSO Do?



- Both Γ point and k-point calculations.
- Both insulators and metals, with smearing.
- Any crystal structure or supercell form.
- Norm conserving pseudopotentials, ultrasoft PPs, PAW.
- LDA, GGA, DFT+U, hybrid functionals, exact exchange, meta GGA, van der Waals corrected functionals.
- Spin polarized calculations, non-collinear magnetism, spin-orbit interactions.
- Nudged elastic band to find saddle points.
- And much more!



II. Doing a “Total Energy” Calculation with the PWscf Package of QE: The SCF Loop

The Kohn-Sham problem

- Want to solve the Kohn-Sham equations:

$$\left[-\frac{1}{2} \nabla^2 + V_{nuc}(\mathbf{r}) + V_H[n(\mathbf{r})] + V_{XC}[n(\mathbf{r})] \right] \psi_i(\mathbf{r}) = \varepsilon_i \psi_i(\mathbf{r})$$

H

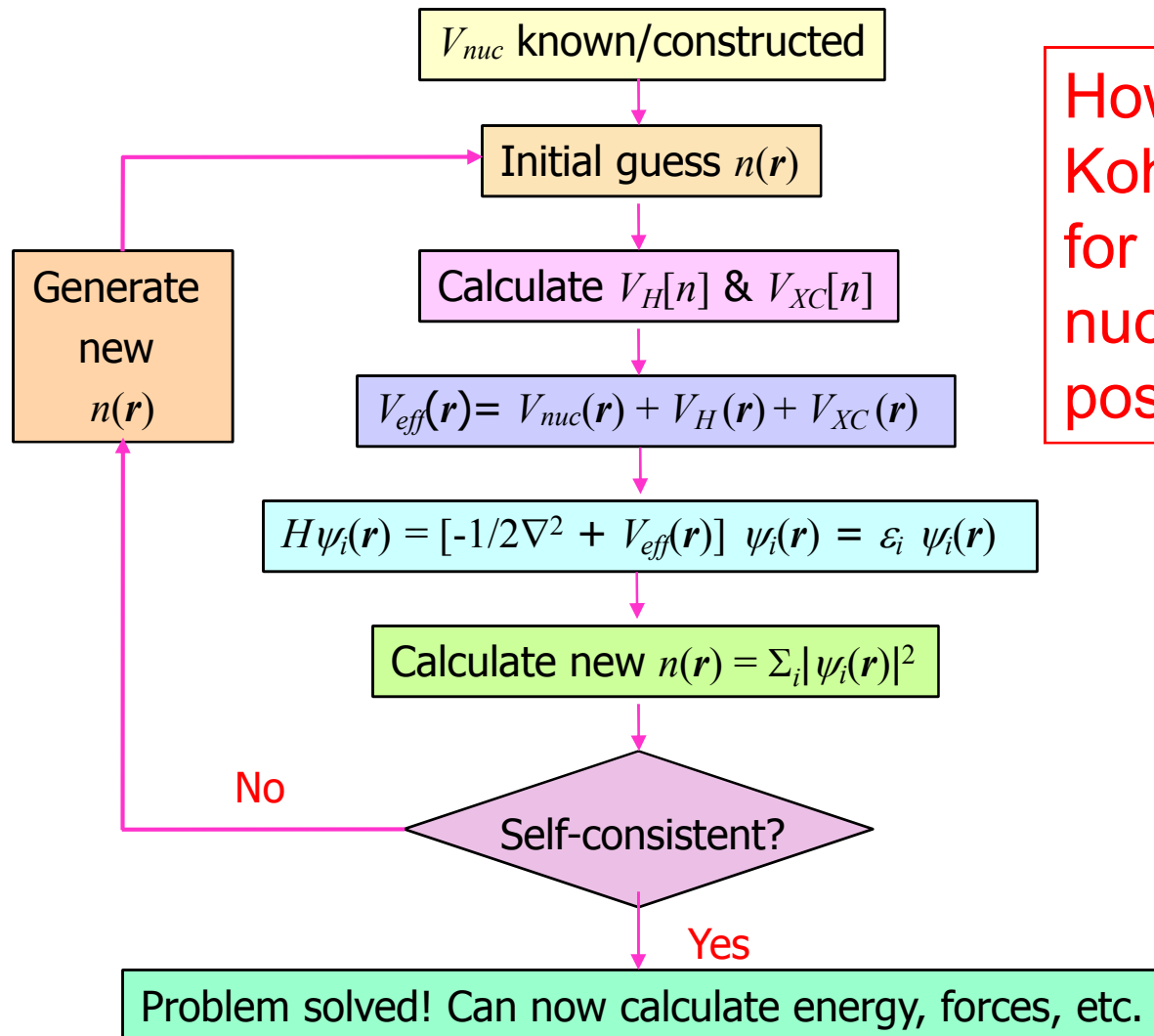
- Note that **self-consistent solution** necessary, as H depends on solution:

$$\{\psi_i\} \rightarrow n(r) \rightarrow H$$

- Convention:

$$e = \hbar = m_e = 1$$

Self-consistent Iterative Solution



How to solve the Kohn-Sham eqns. for a set of fixed nuclear (ionic) positions.

Kohn-Sham Equations in a Basis

- Can choose to expand wavefunctions in a **basis set**:

$$\psi_i(\mathbf{r}) = \sum_{\alpha=1}^{N_b} c_{i\alpha} f_{\alpha}(\mathbf{r})$$

- Eigenvalue equation then becomes:

$$\sum_{\beta} H_{\alpha\beta} c_{i\beta} = \epsilon_i c_{i\alpha}$$

Matrix element *Eigenvalue* *Eigenvector*

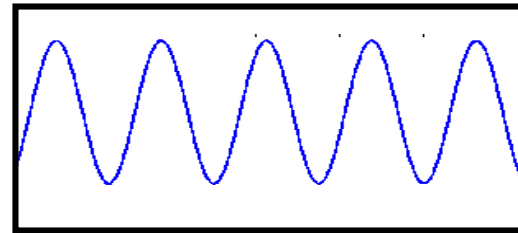
- Solving \Leftrightarrow Have to diagonalize a matrix of size $N_b \times N_b$

Size of basis

Some possible basis sets

- Various possible choices of basis:

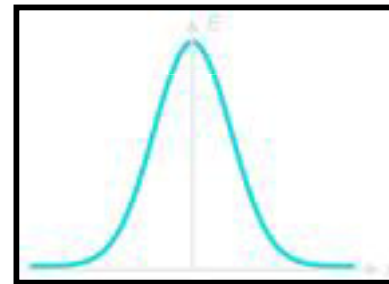
- Plane waves $e^{i\mathbf{K}\cdot\mathbf{r}}$



- Localized sets:

- e.g., Gaussians

- Mixed basis



- Choose so that calculation is fast, accurate, convenient.
- Would like N_b to be small?
- Would like form of $\mathbf{f}_\alpha(\mathbf{r})$ to be simple?

Advantages of a Plane Wave Basis

- **Simple**: Easy to take derivatives, etc. \Rightarrow Coding is easy!
- **Orthonormal**: No overlap integrals.
- **Independent of atomic positions** \Rightarrow No “Pulay forces”; easy to calculate forces for structural relaxation & molecular dynamics.
- **Unbiased**: No assumption about where charge concentrated. (But \therefore also wasteful?)
- **Easy to control convergence** w.r.t. size of basis: only one parameter E_{cut} .
- Can take advantage of FFT's : r-space \leftrightarrow k-space



Disadvantages of a Plane Wave Basis

- The set of plane waves is discrete only if the system is periodic!

(Will discuss...solution = introduction of artificial supercell or periodic approximant.)



- Recall:
 - for free electrons, wavefunction = plane wave.
 - for nearly free electrons, wavefunction = superposition of small number of plane waves.
 - for tightly bound electrons, need a HUGE number of plane waves to get an adequate expansion, i.e., N_b very large!

(Will discuss...solution = introduction of pseudopotentials.)

- Sometimes interpretation harder.

Periodic Systems & Bloch's Theorem

- For a periodic system, recall Bloch's Theorem:

$$\psi_{\mathbf{k}}(\mathbf{r}) = e^{i\mathbf{k} \cdot \mathbf{r}} u_{\mathbf{k}}(\mathbf{r})$$

- $u_{\mathbf{k}}(\mathbf{r})$ has the periodicity of the system, i.e.,

$$u_{\mathbf{k}}(\mathbf{r}) = u_{\mathbf{k}}(\mathbf{r} + \mathbf{R}), \quad \text{where } \mathbf{R} = \text{lattice vector}$$

- As for all lattice-periodic functions, only certain plane waves will appear in the Fourier expansion of $u_{\mathbf{k}}(\mathbf{r})$:

$$u_{\mathbf{k}}(\mathbf{r}) = \frac{1}{\Omega} \sum_{\mathbf{G}} c_{\mathbf{k}, \mathbf{G}} e^{i\mathbf{G} \cdot \mathbf{r}} \quad \text{where } \mathbf{G} = \text{reciprocal lattice vector}$$

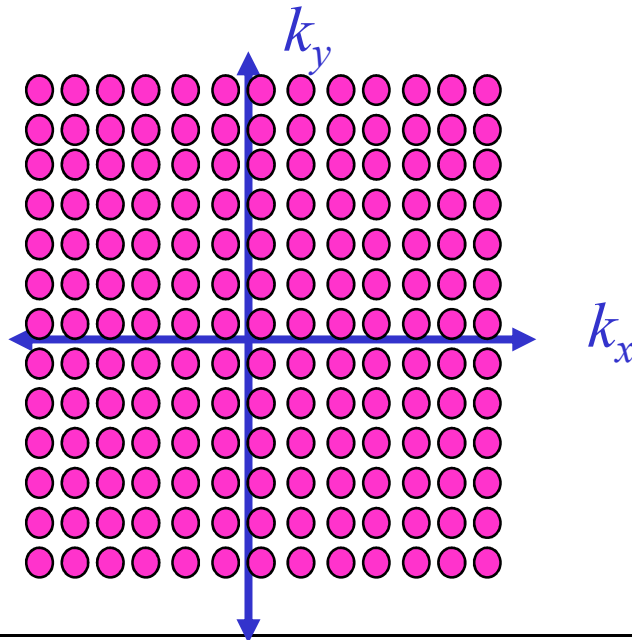
Plane Waves & Periodic Systems

- For a periodic system:

$$\psi_{\mathbf{k}}(\mathbf{r}) = \frac{1}{\Omega} \sum_{\mathbf{G}} c_{\mathbf{k},\mathbf{G}} e^{i(\mathbf{k}+\mathbf{G})\cdot\mathbf{r}}$$

where \mathbf{G} = reciprocal lattice vector

- The plane waves that appear in this expansion can be represented as a grid in k-space:



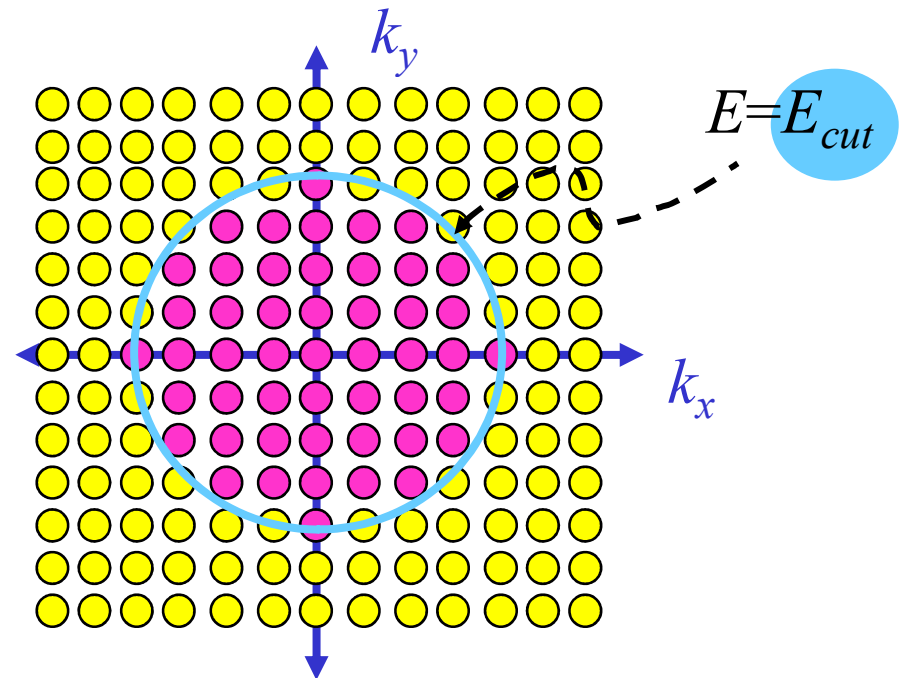
- Only true for periodic systems that grid is discrete.
- In principle, still need infinite number of plane waves.

Truncating the Plane Wave Expansion

- In practice, the contribution from higher Fourier components (large $|\mathbf{k}+\mathbf{G}|$) is small.
- So truncate the expansion at some value of $|\mathbf{k}+\mathbf{G}|$.
- Traditional to express this cut-off in energy units:

$$\frac{\hbar^2 |\mathbf{k} + \mathbf{G}|^2}{2m} \leq E_{cut}$$

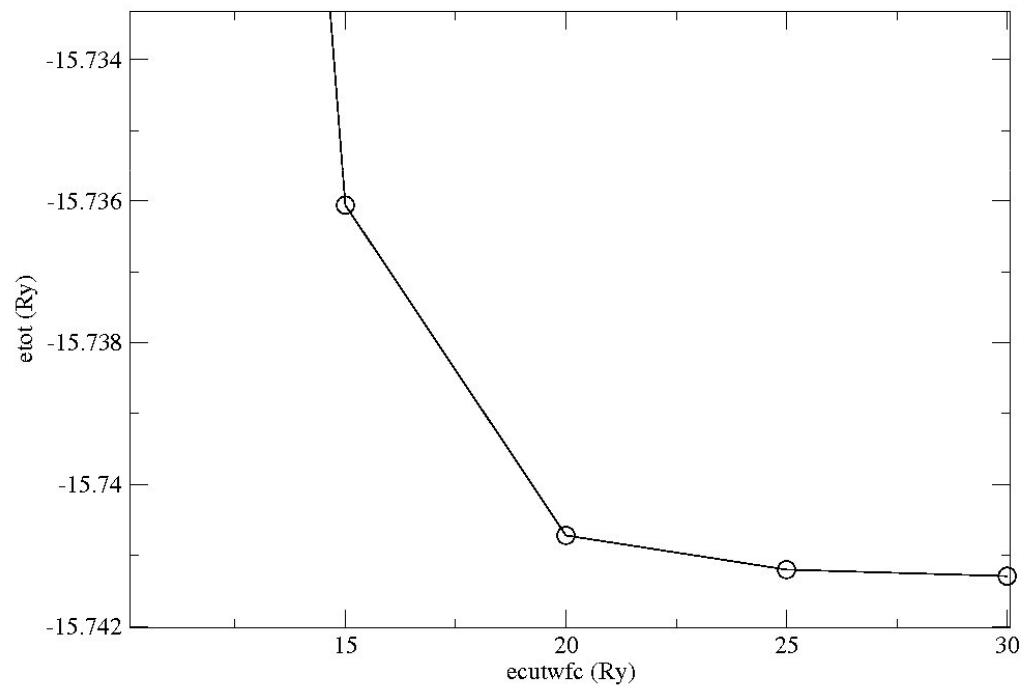
Input parameter **ecutwfc**



Checking Convergence wrt $ecutwfc$

- Must always check.
- Monotonic (variational).

Silicon: Convergence wrt plane wave cutoff



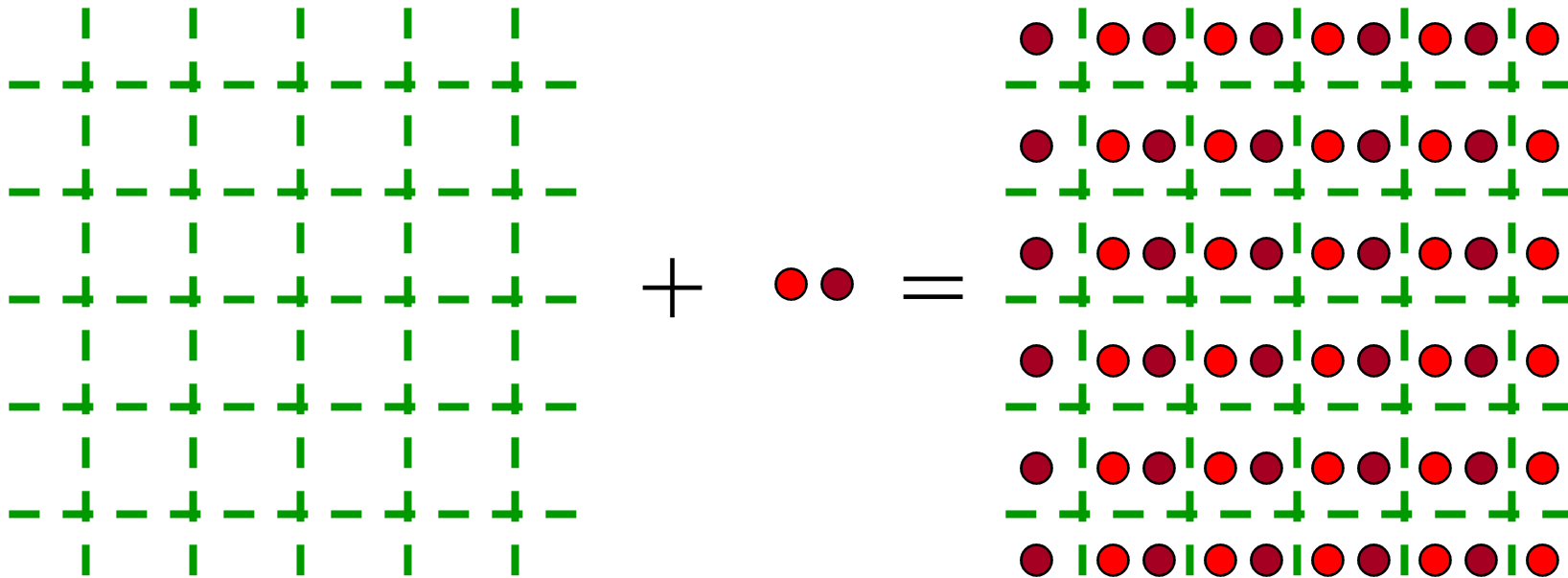
Step 0: Defining the (periodic) system



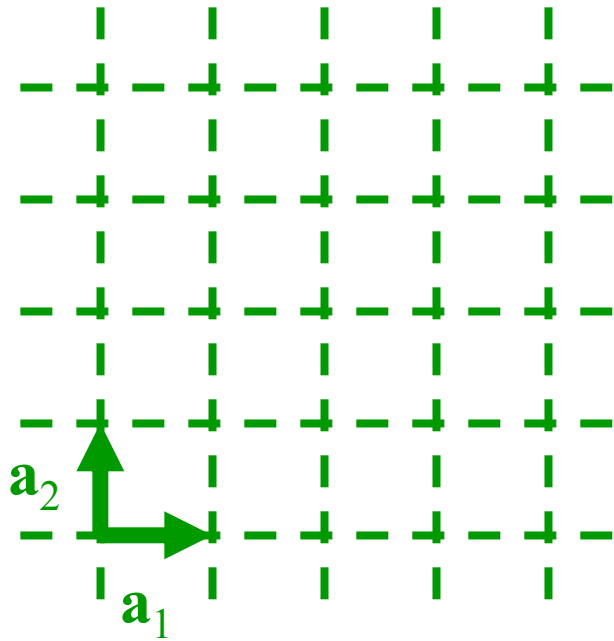
Namelist '**SYSTEM**'

How to Specify the System

- All periodic systems can be specified by a **Bravais Lattice** and an **atomic basis**.



How to Specify the Bravais Lattice / Unit Cell



Input parameter **ibrav**

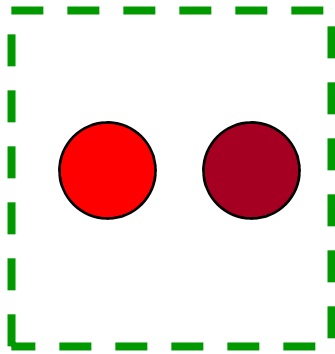
- Gives the type of **Bravais lattice** (SC, BCC, Hex, etc.)

Input parameters {**celldm(i)**}

- Give the lengths [& directions, if necessary] of the lattice vectors a_1 , a_2 , a_3

- Note that one can choose a non-primitive unit cell (e.g., 4 atom SC cell for FCC structure).

Atoms Within Unit Cell – How many, where?



Input parameter **nat**

- Number of atoms in the unit cell

Input parameter **ntyp**

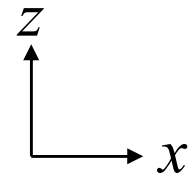
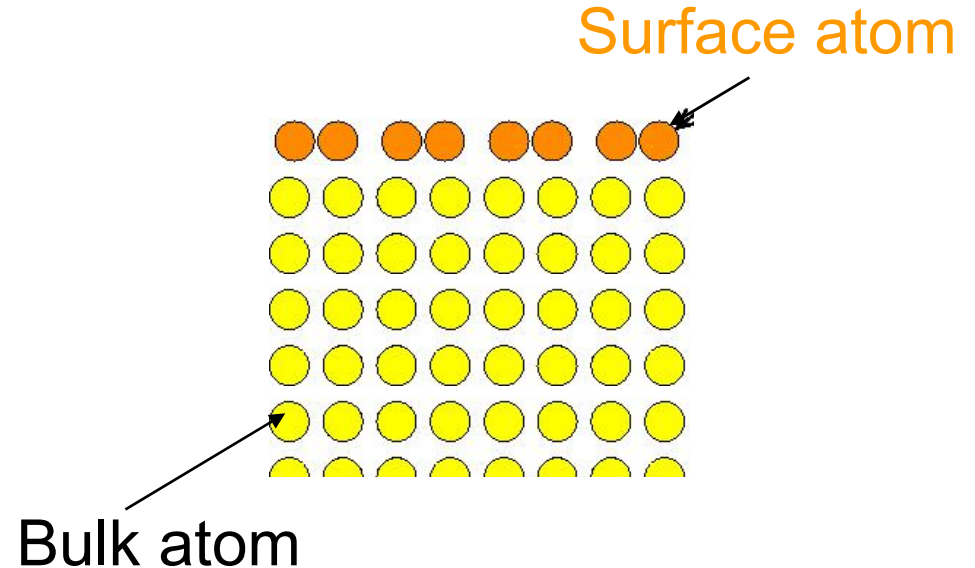
- Number of types of atoms

Card: **ATOMIC_POSITIONS**

- Initial positions of atoms (may vary when “**relax**” done).
- Can choose to give in units of lattice vectors (“**crystal**”) or in Cartesian units (“**alat**” or “**bohr**” or “**angstrom**”)

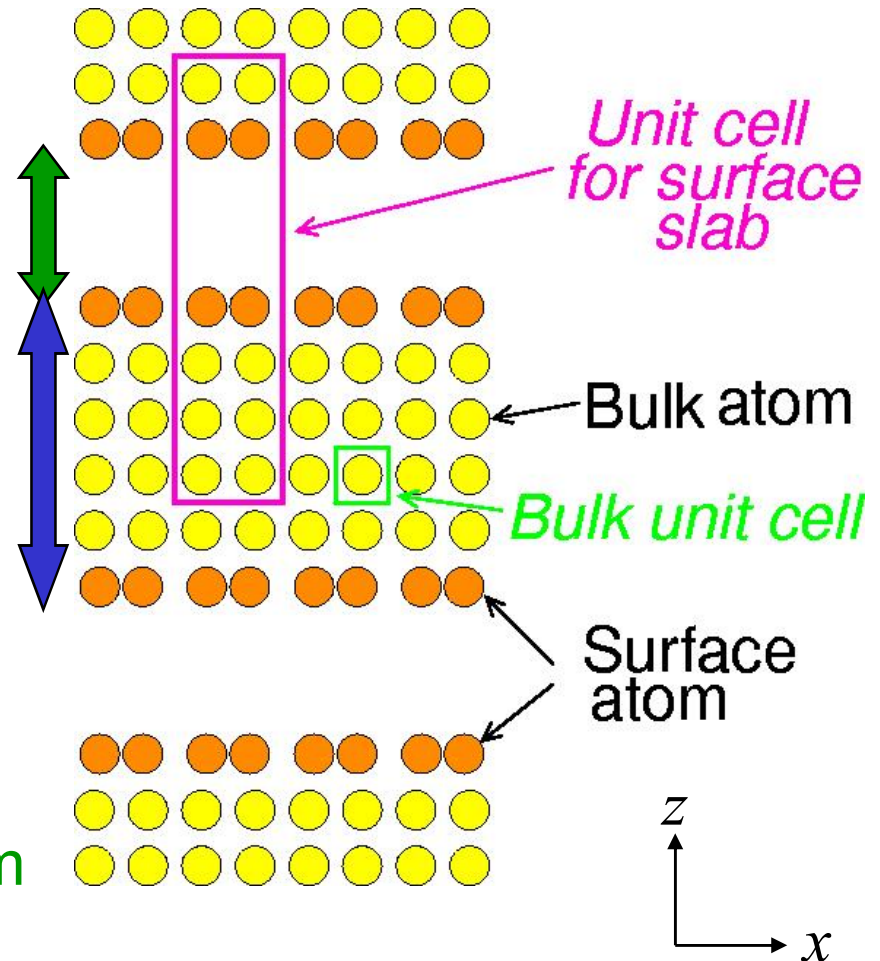
What if the system is not periodic?

- Example 1: Want to study properties of a system with a **surface**.
- Presence of surface \Rightarrow No periodicity along z .



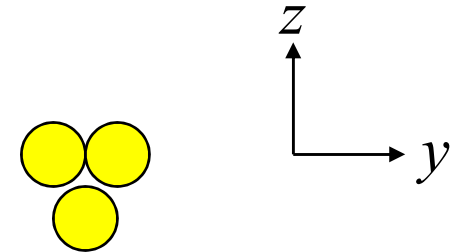
What if the system is not periodic?

- Example 1: Want to study properties of a system with a **surface**.
- Presence of surface \Rightarrow No periodicity along z .
- Use a **supercell**: **artificial periodicity along z** by repeating slabs separated by **vacuum**.
- Have to check convergence w.r.t. **slab thickness** & **vacuum thickness**.



What if the system is not periodic?

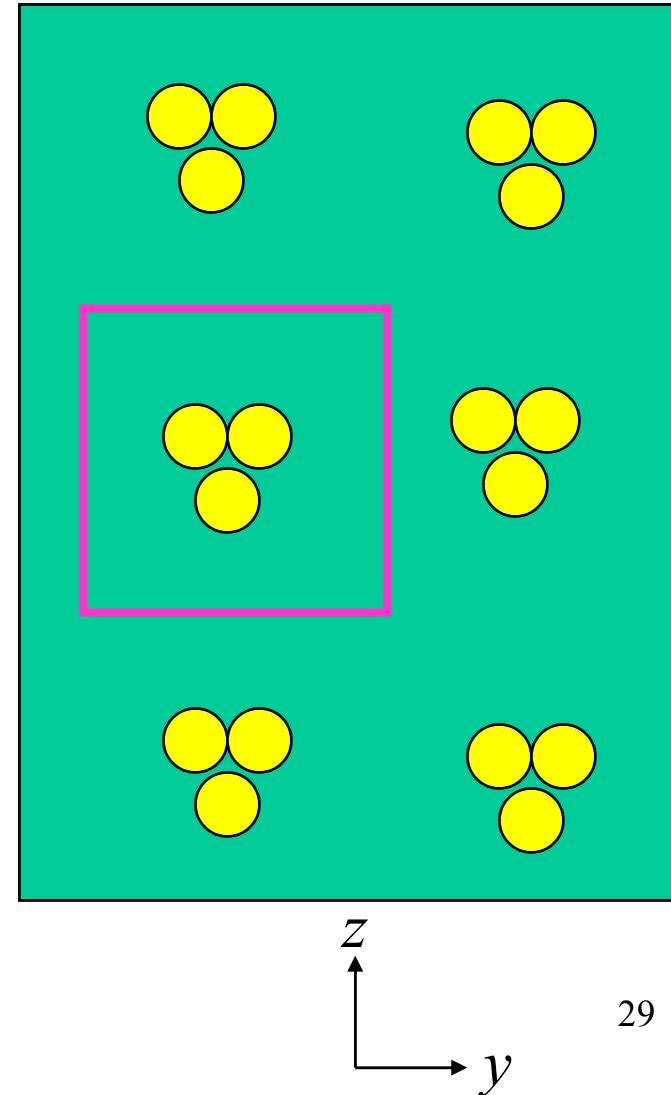
- Example 2: Want to study properties of a nanowire.



- Example 3: Want to study properties of a cluster

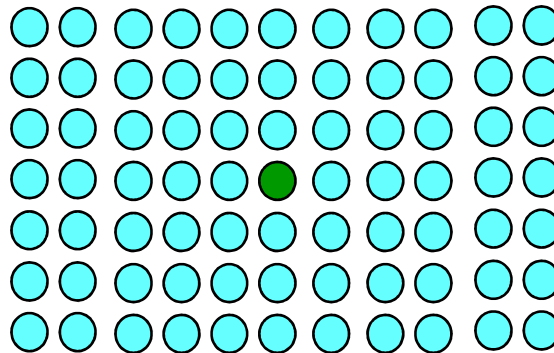
What if the system is not periodic?

- Example 2: Want to study properties of a nanowire \Rightarrow introduce **artificial periodicity along y & z** .
- Example 3: Want to study properties of a cluster \Rightarrow introduce **artificial periodicity along x , y & z** .



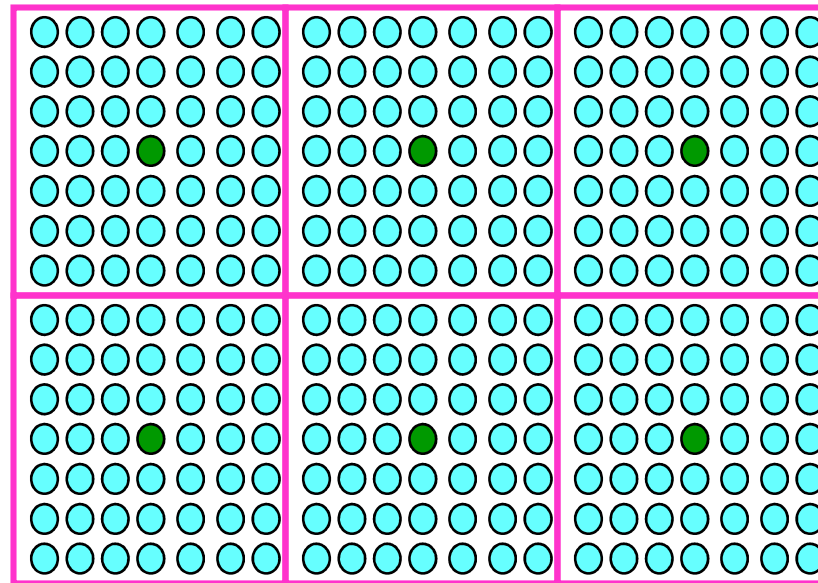
What if the system is not periodic?

- Example 4: Want to study a system with a defect, e.g., a **vacancy or impurity**:



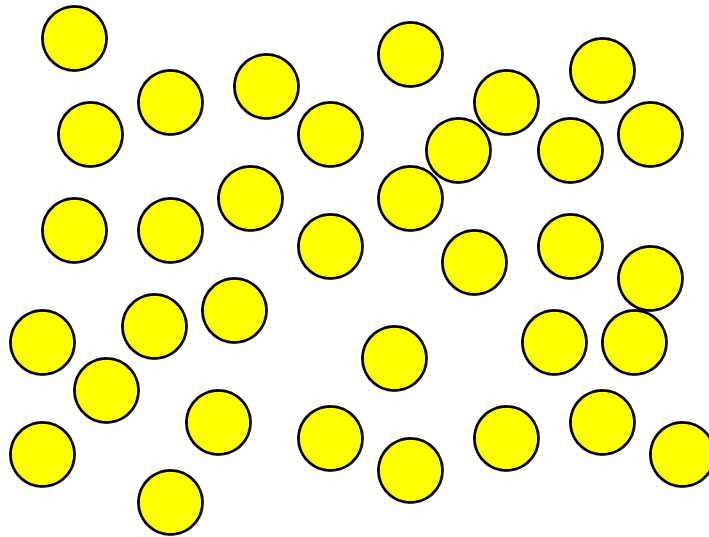
What if the system is not periodic?

- Example 4: Want to study a system with a defect, e.g., a **vacancy or impurity**:



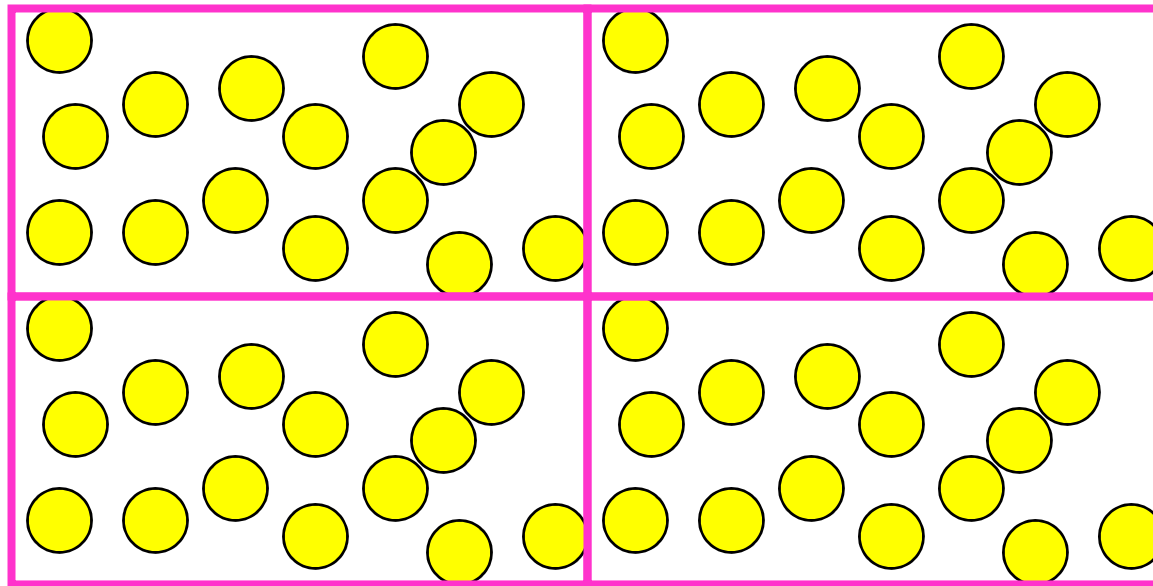
What if the system is not periodic?

- Example 5: Want to study an amorphous or quasicrystalline system.



What if the system is not periodic?

- Example 5: Want to study an amorphous or quasicrystalline system: approximate by a periodic system (with large **unit cell**).



Artificially Periodic Systems \Rightarrow Large Unit Cells

- Note: In all these cases, to minimize the effects of the artificially introduced periodicity, need a large unit cell.



- Long $\mathbf{a}_1, \mathbf{a}_2, \mathbf{a}_3$ (primitive lattice vectors)

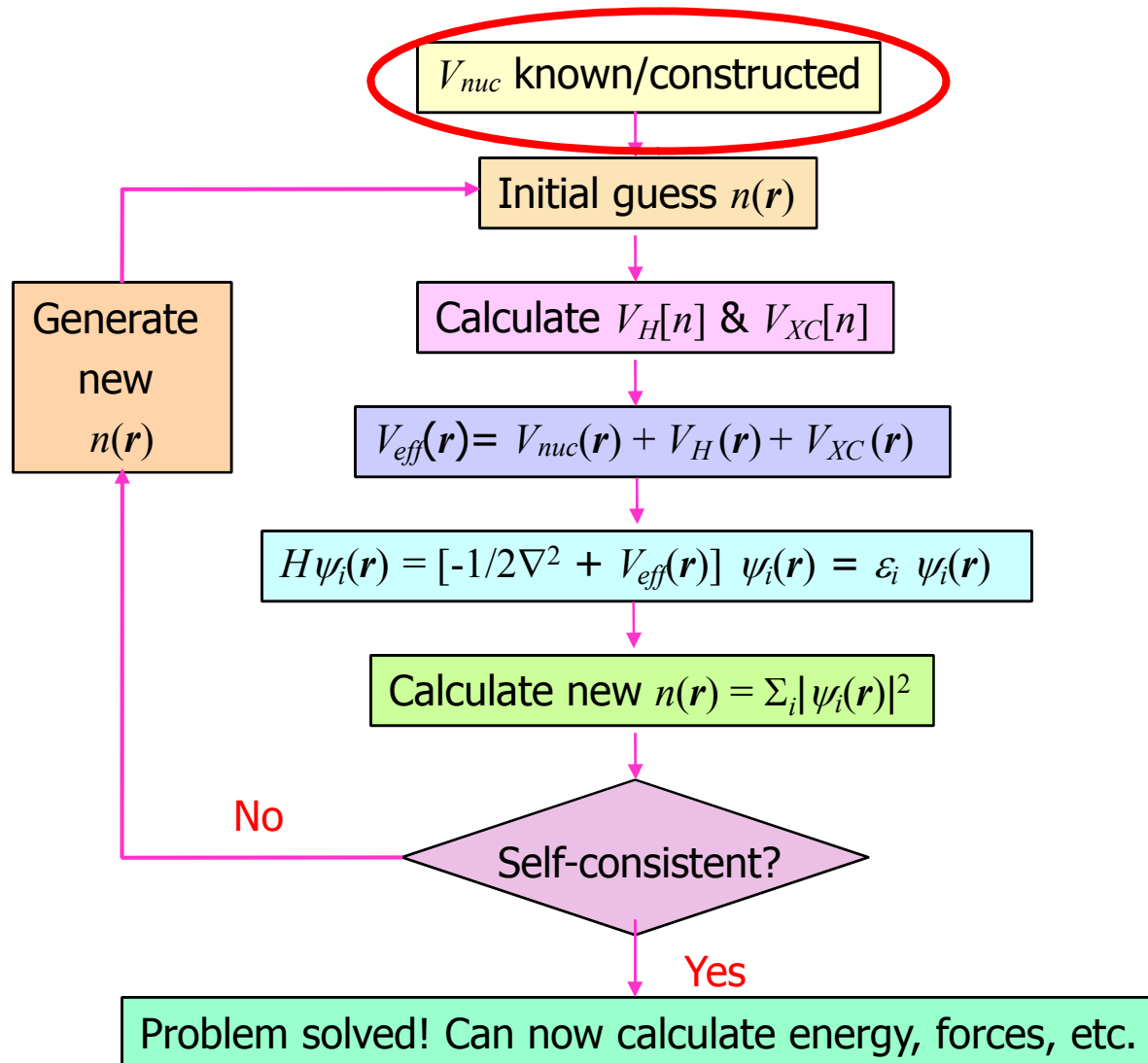


- Short $\mathbf{b}_1, \mathbf{b}_2, \mathbf{b}_3$ (primitive reciprocal lattice vectors)



- Many \mathbf{G} 's will fall within E_{cut} sphere!

Step 1: Obtaining V_{nuc}



Nuclear Potential

- Electrons experience a **Coulomb potential** due to the nuclei.
- This has a known and simple form:

$$V_{nuc} = -\frac{Z}{r}$$

- But this leads to computational problems!

Problem for Plane-Wave Basis

Core wavefunctions:
sharply peaked near
nucleus.

Valence wavefunctions:
lots of wiggles near
nucleus.

High Fourier components present

i.e., need large E_{cut} ☹️

Solutions for Plane-Wave Basis

Core wavefunctions:
sharply peaked near
nucleus.

Valence wavefunctions:
lots of wiggles near
nucleus.

High Fourier components present

i.e., need large E_{cut} ☹️

Don't solve for the
core electrons!

Remove wiggles from
valence electrons.

Pseudopotentials



- Replace nuclear potential by **pseudopotential**
- This is a numerical trick that solves these problems
- There are different kinds of pseudopotentials (Norm conserving pseudopotentials, ultrasoft pseudopotentials, etc.)
- Which kind you use depends on the element.

Will discuss in much greater detail on July 1 (tomorrow).

Pseudopotentials for Quantum Espresso - 1

- Go to <http://www.quantum-espresso.org>; Click on “PSEUDOPOTENTIALS”



QUANTUM ESPRESSO

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NEWS

16.06.14

THE QUANTUM ESPRESSO PRIZE

The Quantum ESPRESSO Foundation, in collaboration with Eurotech, announces the establishment of the *Quantum ESPRESSO prize for quantum mechanical materials modeling*. The prize, which consists of a diploma and a check of one thousand euros, will be awarded annually in January to recognize outstanding doctoral thesis research in the field of quantum mechanical



Zn-induced structural aggregation patterns of β -amyloid peptides. Metallomics, 2012 Courtesy of G. La Penna

QUANTUM ESPRESSO

is an integrated suite of Open-Source computer codes for electronic-structure calculations and

Pseudopotentials for Quantum Espresso - 2

- Click on element for which pseudopotential wanted.

ANY PP LIBRARY		OTHER OPTIONS																			
1 H																	2 He				
3 Li	4 Be															5 B	6 C	7 N	8 O	9 F	10 Ne
11 Na	12 Mg															13 Al	14 Si	15 P	16 S	17 Cl	18 Ar
19 K	20 Ca	21 Sc	22 Ti	23 V	24 Cr	25 Mn	26 Fe	27 Co	28 Ni	29 Cu	30 Zn	31 Ga	32 Ge	33 As	34 Se	35 Br	36 Kr				
37 Rb	38 Sr	39 Y	40 Zr	41 Nb	42 Mo	43 Tc	44 Ru	45 Rh	46 Pd	47 Ag	48 Cd	49 In	50 Sn	51 Sb	52 Te	53 I	54 Xe				
55 Cs	56 Ba	57-70 A	71 Lu	72 Hf	73 Ta	74 W	75 Re	76 Os	77 Ir	78 Pt	79 Au	80 Hg	81 Tl	82 Pb	83 Bi	84 Po	85 At	86 Rn			
87 Fr	88 Ra	89-102 A A	103 Lr	104 Rf	105 Db	106 Sg	107 Bh	108 Hs	109 Mt												
A Lanthanoids			57 La	58 Ce	59 Pr	60 Nd	61 Pm	62 Sm	63 Eu	64 Gd	65 Tb	66 Dy	67 Ho	68 Er	69 Tm	70 Yb					
A A Actinoids			89 Ac	90 Th	91 Pa	92 U	93 Np	94 Pu	95 Am	96 Cm	97 Bk	98 Cf	99 Es	100 Fm	101 Md	102 No					

Upload a file
Download all Files

Pseudopotentials for Quantum-ESPRESSO

Pseudopotential's name
gives information about :

- type of exchange-correlation functional
- type of pseudopotential
- e.g.:

`O.pbe-rrkjus.UPF`

Pseudopotential type: ULTRASOFT
Method: Rappe Rabe Kaxiras Joannopoulos
Functional type: Perdew-Burke-Ernzerhof (PBE) exch-corr
scalar relativistic

Origin: Original QE PP library
Author: Andrea Dal Corso
Generated by Andrea Dal Corso code (rrkj3)
Uploaded by Layla Martin-Samos
Classification controlled by Paolo Giannozzi

`O.pbe-van_ak.UPF`

Pseudopotential type: ULTRASOFT
Method: Vanderbilt ultrasoft
Functional type: Perdew-Burke-Ernzerhof (PBE) exch-corr
scalar relativistic

Origin: Original QE PP library
Generated by Vanderbilt code version 7.3.4
More Information: `O.pbe-van_ak.txt`
Uploaded by Layla Martin-Samos
Classification controlled by Paolo Giannozzi

`O.pbe-rrkjus.UPF` (details)

→ Perdew-Burke-Ernzerhof (PBE) exch-corr

→ Rabe Rappe Kaxiras Joannopoulos (ultrasoft)

Element & V_{ion} for Quantum-ESPRESSO

e.g, for calculation on $BaTiO_3$:

ATOMIC_SPECIES

Ba 137.327 Ba.pbe-nsp-van.UPF

Ti 47.867 Ti.pbe-sp-van_ak.UPF

O 15.999 O.pbe-van_ak.UPF

- **ecutwfc**, **ecutrho** depend on type of pseudopotentials used (should test).
- When using ultrasoft pseudopotentials, set **ecutrho** = $8-12 \times \text{ecutwfc}$!!

Element & V_{ion} for Quantum-ESPRESSO

- Should have same exchange-correlation functional for all pseudopotentials.

input

```
diagonalization = cg, mixing_beta = 0.7, conv_thr = 1.0
/
ATOMIC_SPECIES
Fe 55.85 Fe pz-nd-rrkjus.UPF
Co 58.93 Co pbe-nd-rrkjus.UPF
ATOMIC_POSITIONS (crystal)
Fe 0.00 0.00 0.00
```

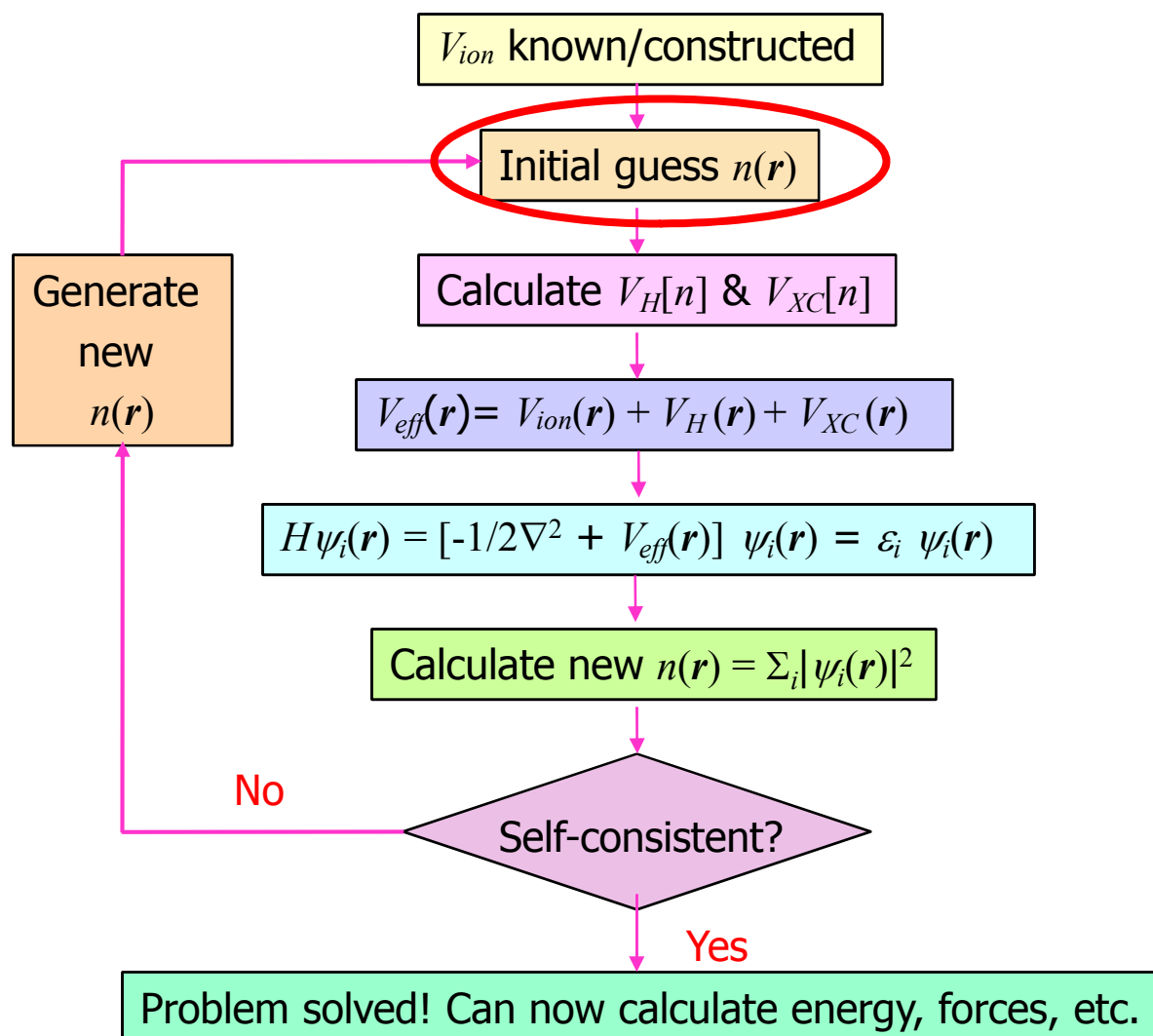
oops!

output

```
Max angular momentum in pseudopotentials
%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%
from readpp : error #          2
inconsistent DFT read
%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%%

stopping ..█
```


Step 2: Initial Guess for $n(\mathbf{r})$



Starting Wavefunctions

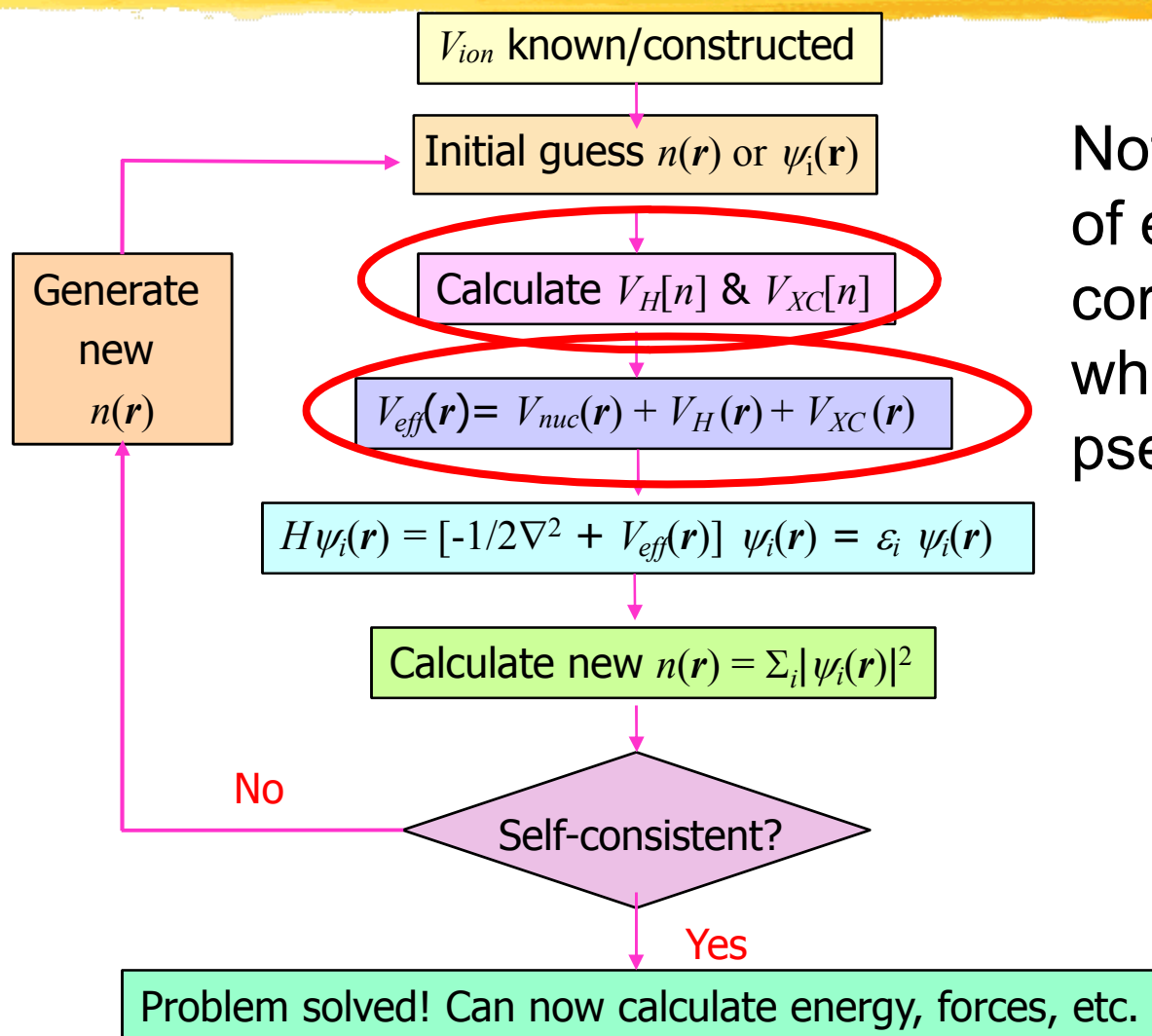
The closer your starting wavefunction is to the true wavefunction (which, of course, is something you don't necessarily know to start with!), the fewer the scf iterations needed.

```
startingwfc 'atomic'  
            'atomic+random'  
            'random'  
            'file'
```

Superposn of atomic orbitals

“The beginning is the most important part of the work” - Plato

Steps 3 & 4: Effective Potential



Note that type of exchange-correlation chosen while specifying pseudopotential



Exchange-Correlation Potential

- $V_{xc} \equiv \delta E_{xc} / \delta n$ contains all the many-body information.
- Known [numerically, from Quantum Monte Carlo ; various analytical approximations] for **homogeneous electron gas**.

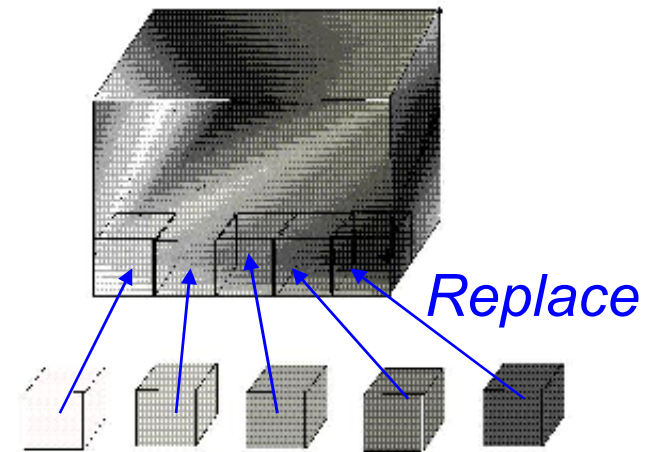
- Local Density Approximation:

$$E_{xc}[n] = \int n(\mathbf{r}) V_{xc}^{\text{HOM}}[n(\mathbf{r})] d\mathbf{r}$$

-surprisingly successful!

pz

(in name of pseudopotential)

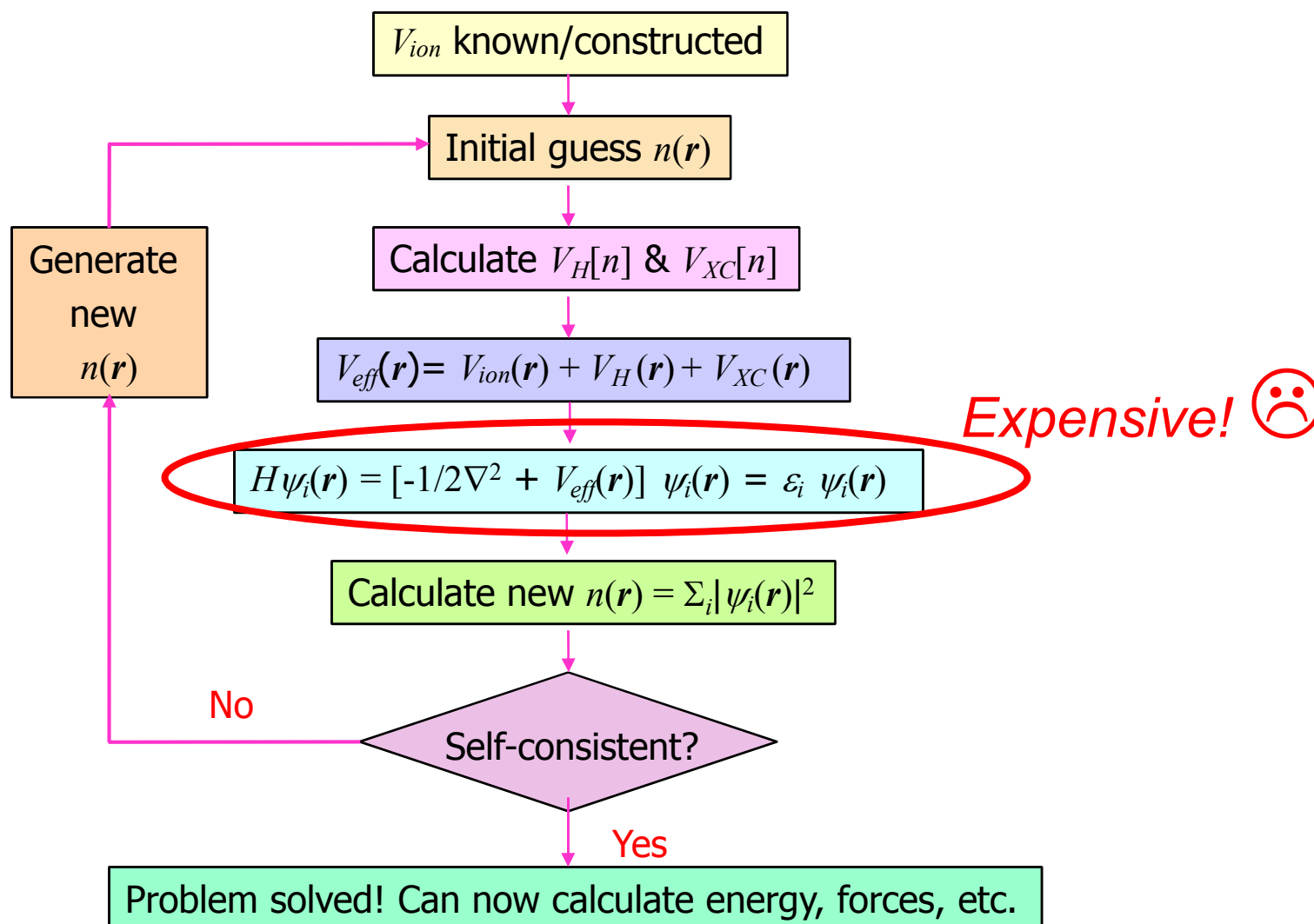


- Generalized Gradient Approximation(s): Include terms involving gradients of $n(\mathbf{r})$

pw91, pbe

(in name of pseudopotential)

Step 5: Diagonalization



Diagonalization

- Need to diagonalize a matrix of size $N_{PW} \times N_{PW}$
- $N_{PW} \gg N_b$ = number of bands required = $N_e/2$ or a little more (for metals).
- OK to obtain lowest few eigenvalues.
- Exact diagonalization is expensive!
- Use **iterative diagonalizers** that recast diagonalization as a minimization problem.

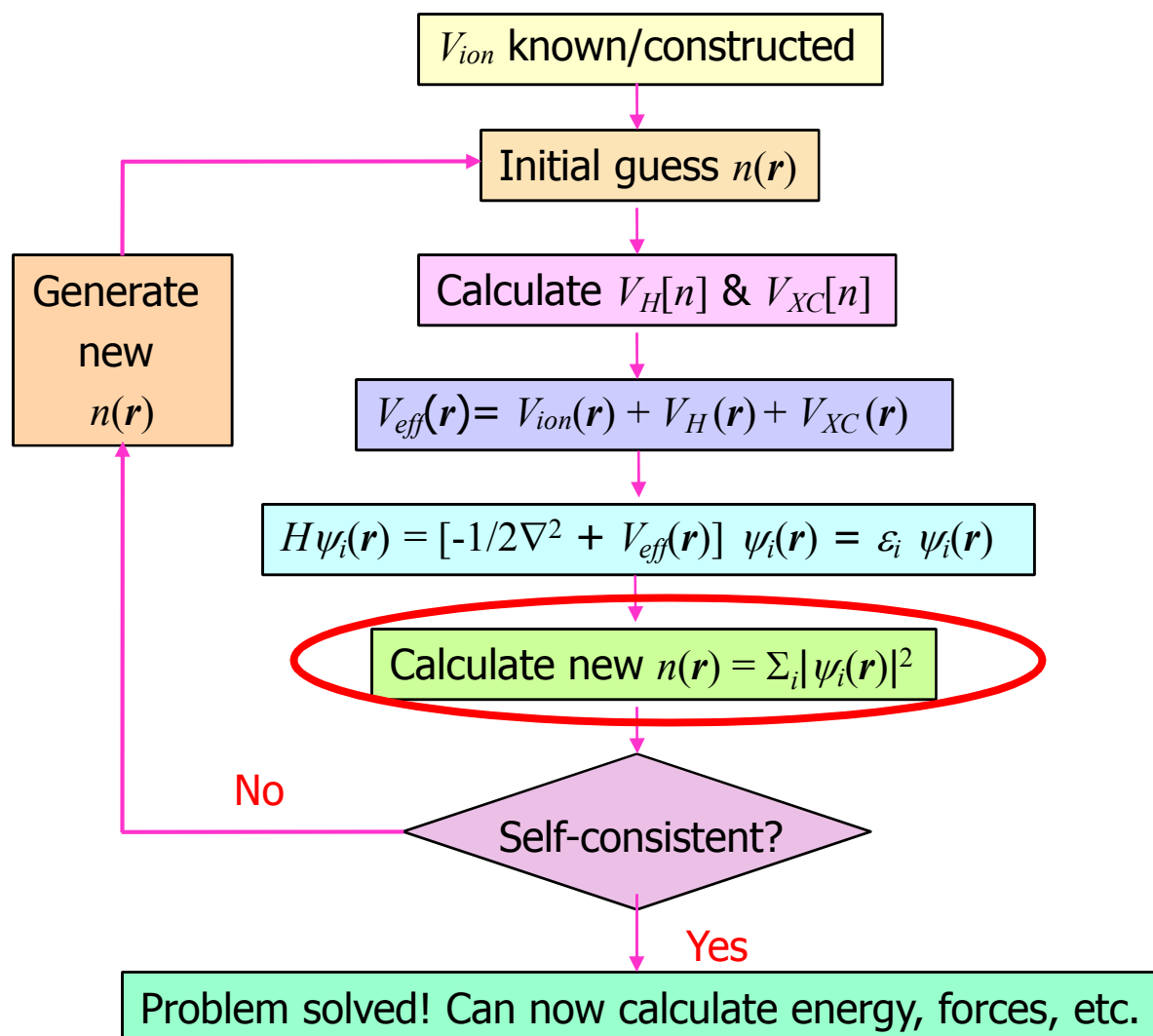
Input parameter **diagonalization**

-which algorithm used for iterative diagonalization

Input parameter **nbnd**

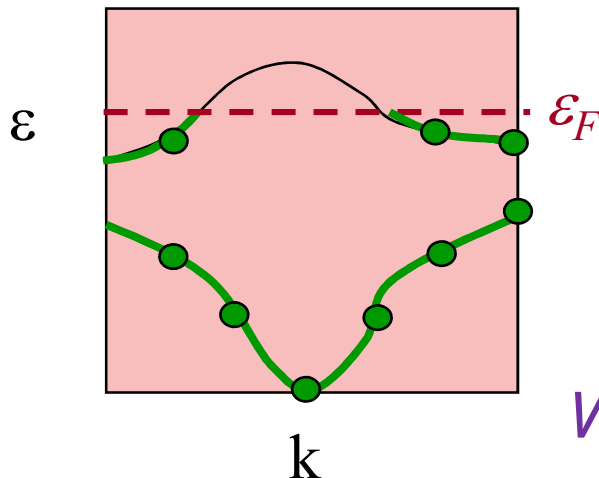
*-how many eigenvalues computed
for metals, choose depending on value of **degauss***

Step 6: New Charge Density



Brillouin Zone Sums

- Many quantities (e.g., n , E_{tot}) involve sums over \mathbf{k} .
- In principle, need infinite number of \mathbf{k} 's.
- In practice, sum over a finite number: BZ “Sampling”.
- Number needed depends on band structure.
- Typically need more \mathbf{k} 's for metals.
- Need to test convergence wrt k-point sampling.

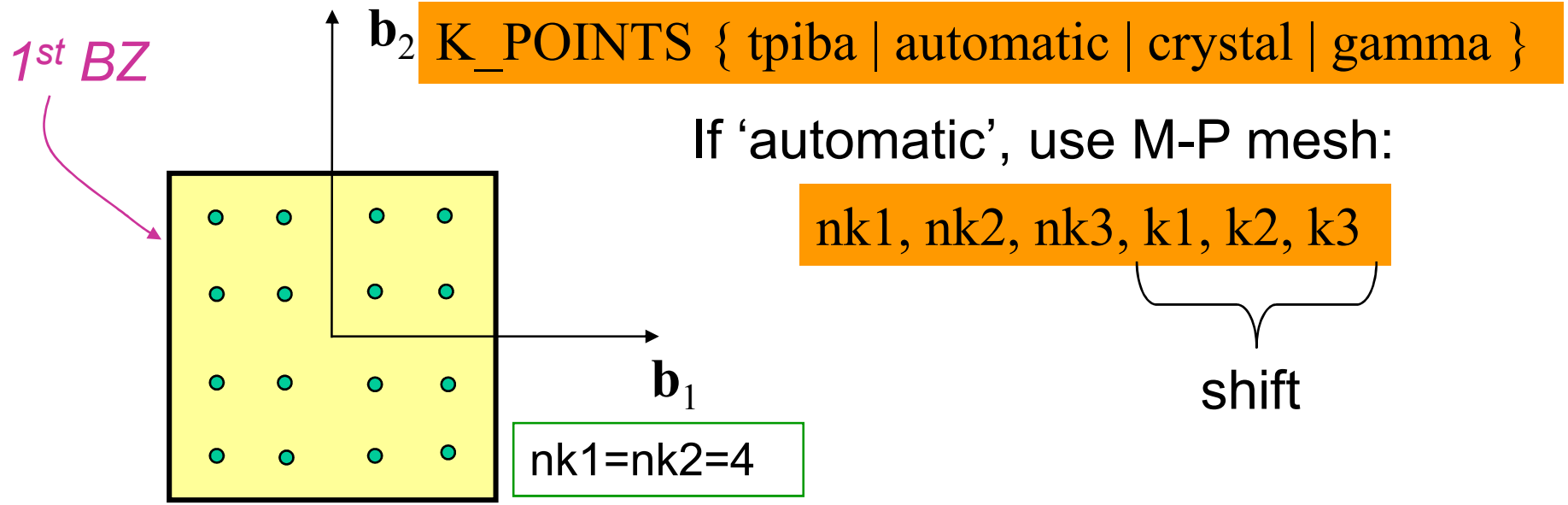


$$\langle P \rangle = \frac{1}{N_{\mathbf{k}}} \sum_{\mathbf{k} \in BZ} P(\mathbf{k}) w_{\mathbf{k}}$$

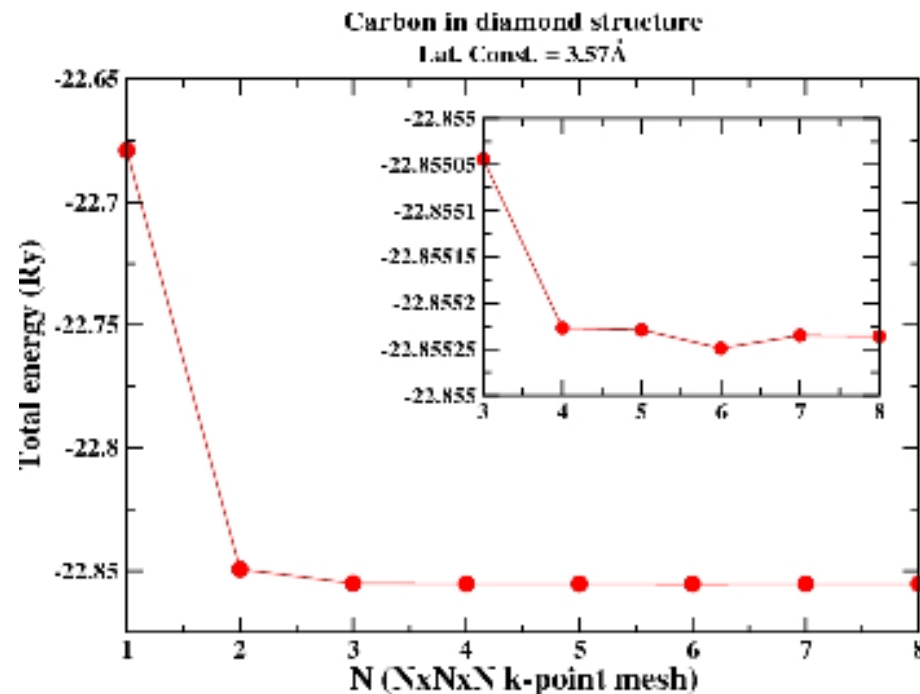
Will discuss in greater detail tomorrow

Types of k-point meshes

- **Special Points:** *[Chadi & Cohen]*
Points designed to give quick convergence for particular crystal structures.
- **Monkhorst-Pack:**
Equally spaced mesh in reciprocal space.
May be centred on origin ['non-shifted'] or not ['shifted']



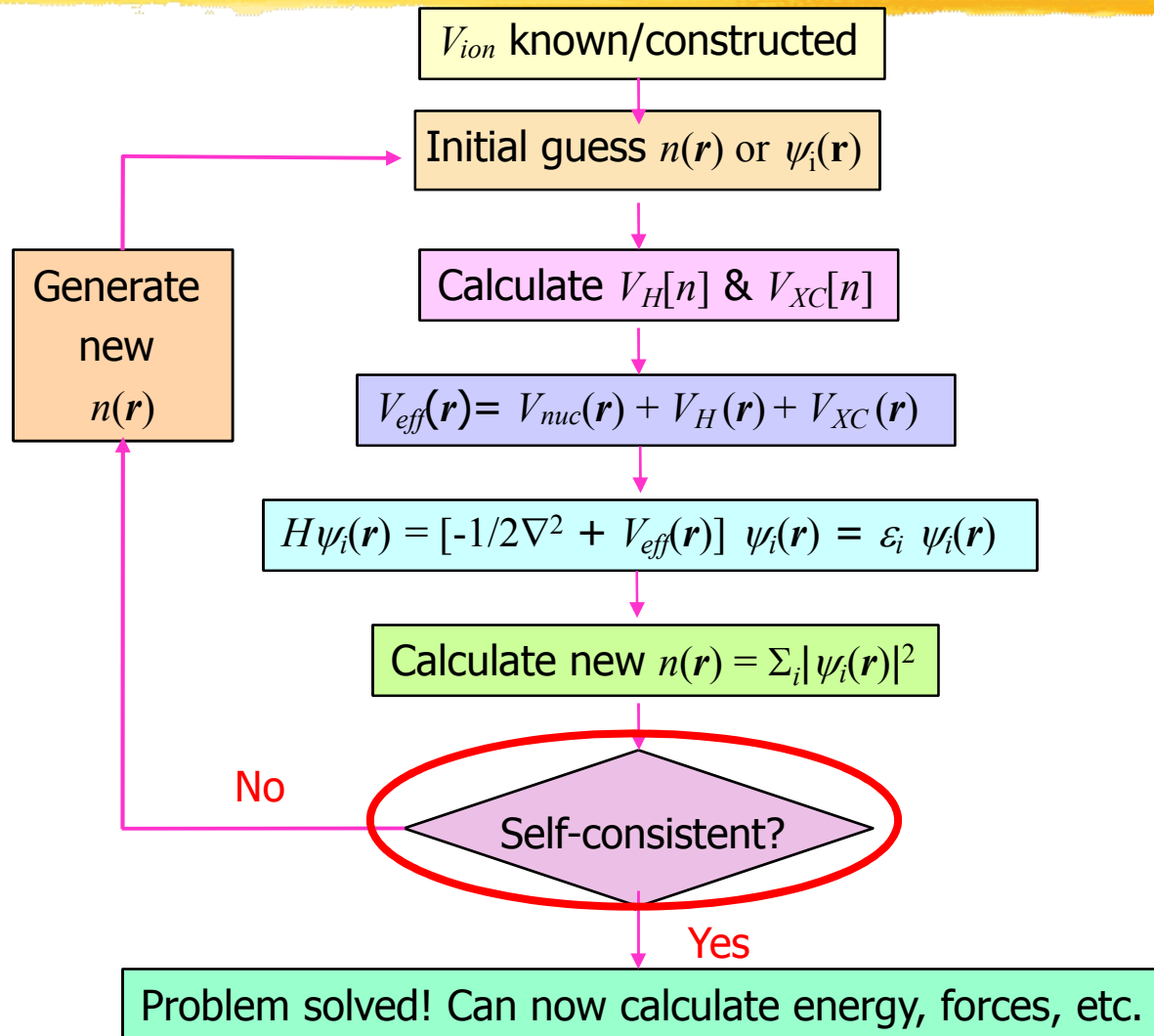
Convergence wrt BZ sampling



Madhura Marathe

Note: Differences in energy usually converge faster than absolute value of total energy because of error cancellation (if supercells & k-points are identical or commensurate).

Step 7: Check if Convergence Achieved



Testing for scf convergence

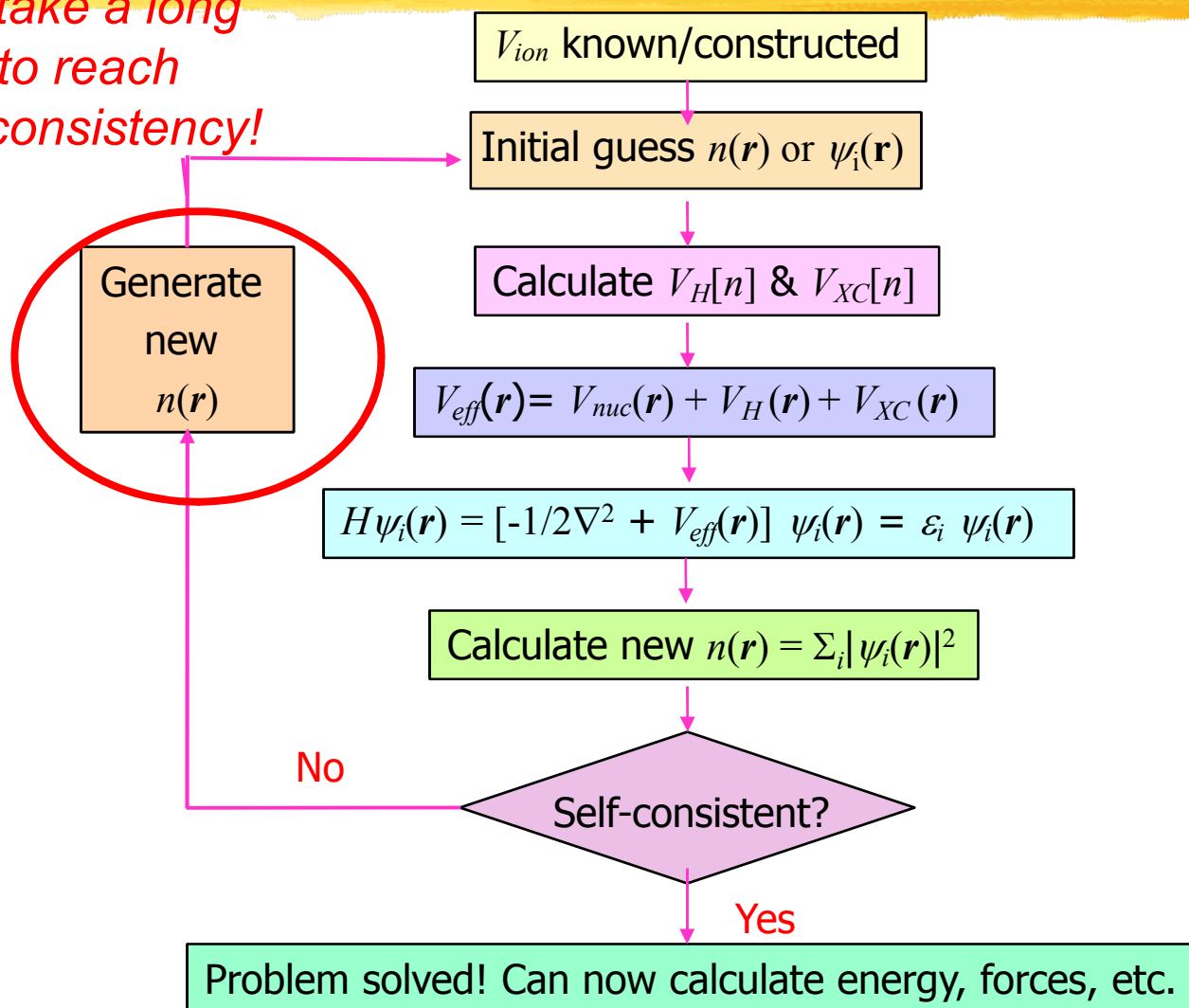
- Compare n th and $(n-1)$ th approximations for density, and see if they are close enough that self-consistency has been achieved.



Input parameter **conv_thr**

Step 8: Mixing

Can take a long time to reach self-consistency!



Mixing

- Iterations n of self-consistent cycle:
- Successive approximations to density:
 $n_{in}(n) \rightarrow n_{out}(n) \rightarrow n_{in}(n+1)$.
- $n_{out}(n)$ fed directly as $n_{in}(n+1)$?? No, usually doesn't converge.
- Need to **mix**, take some combination of input and output densities (may include information from several previous iterations).
- Goal is to achieve **self consistency** ($n_{out} = n_{in}$) in as few iterations as possible.

Mixing in Quantum-ESPRESSO



Input parameter **mixing_mode**

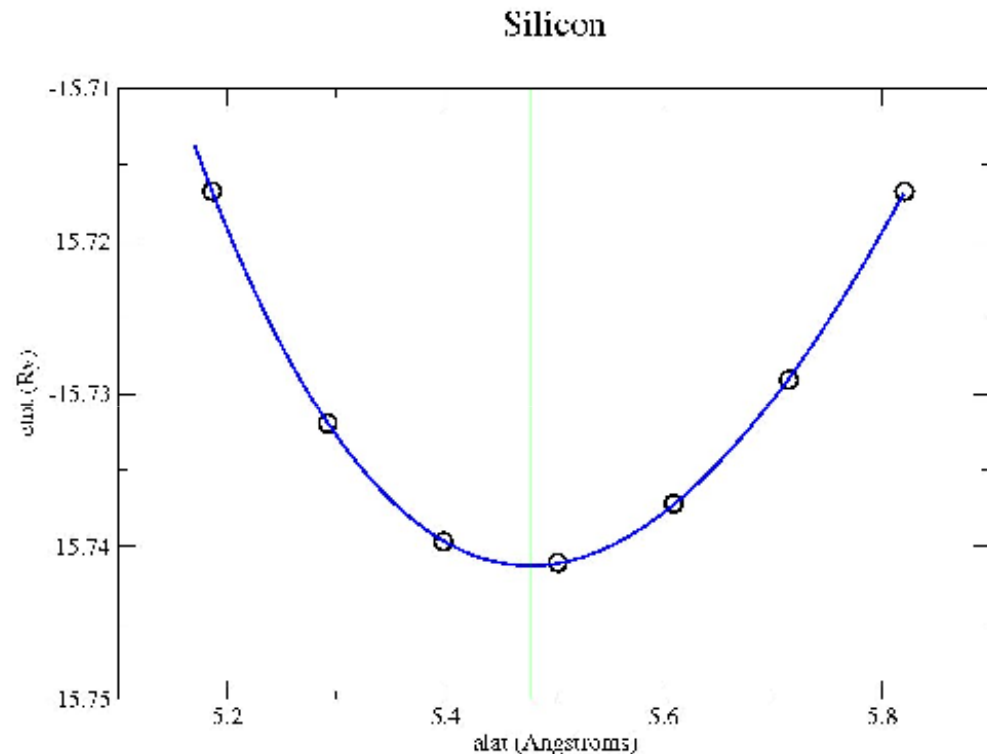
-Prescription used for mixing.

Input parameter **mixing_beta**


- How much of new density is used at each step*
- Typically use value between 0.1 & 0.7*

Output Quantities: Total Energy

- Perhaps the most important output quantity is the TOTAL ENERGY
- Can use, e.g., to optimize structure
- e.g., for a cubic crystal, where the structure can be specified by a single parameter (side of cube):



Energy vs. lattice constant or volume is given by
“Equation of State”



IV. Structure of PWscf Input Files

PWscf input file



- For documentation about input parameters for PWscf, read **INPUT_PW.html** in the **Doc** subdirectory.
- The PWscf input file is structured into **NAMELISTS** and **INPUT_CARDS**.

PWscf NAMELISTs in Input File

- There are three mandatory **NAMELISTs** :
- **&CONTROL** input variables that control the type of calculation performed and the amount of I/O.
- **&SYSTEM** input variables that specify the system.
- **&ELECTRONS** input variables that control the algorithms used to reach a self-consistent solution of the Kohn-Sham equations.
- There are other (optional) namelists...

PWscf CARDS in Input File

- There are three mandatory **CARDS** :
- **ATOMIC_SPECIES** name, mass and pseudopotential used for each species in system.
- **ATOMIC_POSITIONS** coordinates of each atom in unit cell.
- **K_POINTS** coordinates and weights of the k-points used for BZ sums..
- There are other (optional) CARDS...

Other Features / Types of Calculations

- Spin Polarized Calculations (Magnetism)
- Density Functional Perturbation Theory (Phonons)
- Nudged Elastic Band (Barriers)
- Molecular Dynamics
- ...and much, much more!



*It's not a **bird**...*

*It's not **Superman**...*

*It's a **Plane** Wave !*

The End!

Have fun with Quantum-ESPRESSO!

